

Crack Front Dynamics: The Interplay of Singular Geometry and Crack Instabilities

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(Received 11 December 2014; published 1 May 2015)

When fast cracks become unstable to microscopic branching (microbranching), fracture no longer occurs in an effective 2D medium. We follow in-plane crack front dynamics via real-time measurements in brittle gels as microbranching unfolds and progresses. We first show that *spatially local* energy balance quantitatively describes crack dynamics, even when translational invariance is badly broken. Furthermore, our results explain microbranch dynamics; why microbranches form along spatially localized chains and how finite-time formation of cusps along the crack front leads to their death.

DOI: 10.1103/PhysRevLett.114.175501

PACS numbers: 62.20.mm, 46.50.+a, 62.20.mt

Although fracture mechanics is a mature field, many of its simplest questions remain unanswered. Cracks govern material stability. The classic 2D descriptions of fracture [1] show that the existence of a crack focuses elastic energy stored in a strained material into a single point—the crack tip. There, stresses diverge as $\sigma \sim K/\sqrt{r}$, where r is the distance from the tip and K is a coefficient called the stress intensity factor. The motion of the crack is then governed by the physical demand of energy balance, that G , the elastic energy flowing to the crack tip, equals the fracture energy Γ (the energy dissipated per unit area of crack extension). For example, a sudden increase in Γ will cause a corresponding decrease in crack velocity, since part of the elastic energy that is used to separate the crack faces is now used to compensate for the increase in fracture energy. Energy balance is the basis for the continuum description of the dynamics of simple, straight cracks in brittle materials.

Cracks, however, fracture 3D materials. Whereas a crack's tip in 2D materials is a singular *point*, in three dimensions the leading edge of a crack forms a singular *line*, the *crack front*. In “simple” cracks, the crack front is a straight line with no overt dynamics. For this reason, 2D descriptions of fracture dynamics for simple cracks are sufficient and the classical 2D theory delineated above [1] describes crack dynamics perfectly [2–4], as long as they propagate along a straight-line trajectory.

Simple cracks are generically unstable entities [5] that leave complex 3D surfaces in their wake [6–9]. Their intrinsic 3D character calls for a study of the dynamics of crack fronts [10–14]. Even when a crack is constrained to a plane, the geometry of the crack front is important. For example, the advance of quasistatic fronts in patterned materials is influenced by the long-range elastic forces that act along crack fronts [15–17]. A recent study [18] of quasistatic peeling of adhesive tape demonstrates the striking consequences of manipulating front geometry to enhance material toughness. The elasticity of crack fronts is also important to explain the fluctuating dynamics of

quasistatic planar cracks propagating through random heterogeneities [19–21].

When cracks are rapid, experiments suggest [22] that distortions of a crack front produce inertia in the “massless” cracks described by 2D theory, an intrinsic 3D effect. The microbranching instability of rapid simple cracks is an example where a crack front loses its symmetry *dynamically*. Simple cracks in brittle materials experience this instability above a critical crack velocity $v_c \sim 0.3c_R$, where c_R is the material's Rayleigh wave speed [5,22,23]. For velocities $v > v_c$, directed chains (*branch lines*) of micron-scale branched cracks are spontaneously generated [see Fig. 1(b)]. Each “microbranch” branches away from the main crack to propagate beneath its fracture

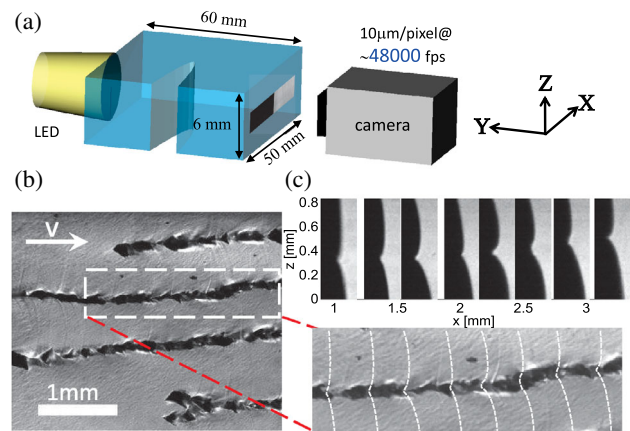


FIG. 1 (color online). (a) The experimental setup. A rectangular block of polyacrylamide gel is strained by displacement of its Y boundaries. Crack fronts, propagating within the sample's mid-plane are visualized via shadowgraph with a high-speed camera; illumination is via collimated LED light normal to the fracture (XZ) plane. (b) A post-mortem fracture surface section containing four nearly parallel directed chains of microbranches (branch lines). (c) Top: The imaged fronts that formed the branch line denoted by the box in (b). Bottom: A close-up photograph of the corresponding fracture surface with overlaid edge-detected fronts.

surface. Every microbranch creates an additional fracture surface and hence, increases the value of Γ that is felt by the main crack [24]. As, for short times, the total energy flux to the crack front is constant, microbranching also reduces the energy flux to the rest of the crack. A single microbranch may effectively increase Γ by up to 100% (if it creates an additional crack propagating parallel to the main crack). One may therefore consider microbranches as energy sinks that are dynamically “toggled” on and off. As microbranches are localized along the crack front, these perturbations are spatially localized in both parallel and transverse to the propagation direction. Therefore, when excited, microbranches locally perturb initially straight crack fronts to produce large fluctuations in v .

In this Letter, we study the propagation of fast crack fronts during the microbranching instability. We will provide a detailed account of in-plane front dynamics as microbranches nucleate, grow, and eventually die. This will provide us with new insights as to how front geometry and motion continuously feed each other, eventually conspiring to cause microbranch “death.”

We study crack front dynamics by performing real-time visualization of the fronts as they traverse the fracture surface. We do this by using brittle polyacrylamide gels composed of a 13.8% (w/v) of acrylamide and a 1:37.5 (w/w) bisacrylamide-to-acrylamide ratio, where $c_R = 5.2$ m/s. When scaled by c_R , these gels exhibit both the same single crack dynamics [4] and microbranching phenomenology [23] as more conventional brittle materials such as glass and PMMA. The advantage of using gels is in reducing the wave speeds, hence crack velocities, by 3 orders of magnitude.

Our experimental system is schematically described in Fig. 1(a). Our gel samples are cast to be rectangular blocks of dimensions $50 \times 60 \times 6$ mm³ ($X \times Y \times Z$), where X , Y , and Z are, respectively, the propagation, loading, and thickness directions. We image the crack front by shining collimated LED light through the samples in a direction normal to the XZ (fracture) plane. The strong curvature at the crack tip deflects the light at the crack front, creating a sharp shadow boundary in the image plane and producing a 2D projection of the crack front. We capture front dynamics by imaging the front using high-speed (IDT-Y4) camera at ~ 48000 frames/sec in a “ $X \times Z$ ” window of 10×1 mm² located at the center section of the gel sample. Our spatial resolution was ~ 10 micron per pixel.

We generated crack front dynamics in the microbranching regime by displacing the system’s boundaries by a 10%–12% strain, prior to initiating fracture. Fracture was initiated by inserting a small “seed” crack at the sample’s edge, midway between its vertical boundaries. This procedure produced crack velocities of 0.1 – $0.5c_R$ along the midplane of the sample. All velocities $v(z)$ referred to in this Letter are the *normal* velocities to the front at each point z .

In Fig. 1(b) we present a typical picture of the fracture surface created. Crack propagation is complex, forming microbranches at several locations along the front. The four branch lines appearing in the figure are oriented parallel to the mean crack velocity. To characterize the corresponding front dynamics, we focus on the formation of a single branch line, denoted by the dashed box in Fig. 1(b). In Fig. 1(c) we present a series of snapshots of successive crack fronts within this box, together with a close-up of the resulting *post-mortem* branch line formed by these fronts. A direct comparison between the fronts and resulting fracture surface [Fig. 1(c), bottom] shows that a one-to-one correspondence exists between microbranches on the fracture surface and loci of high curvature along the front.

Figure 2 presents a detailed picture of the motion within a typical section of a front that includes three well-defined microbranching events. Along the crest of a single branch line marked **A** we can see that, prior to the nucleation of the first microbranch (**1**) the front has nearly zero curvature. Then, with the nucleation of the first microbranch (**2**) the local front velocity along **A** gradually decreases. The apparent cause for the slowing down is the effective increase in local fracture energy due to the incipient microbranching. The resulting gradient in velocity along the front produces an increase of front curvature. The front acquires a locally concave shape that becomes more and more pronounced until (**3**) the curvature suddenly drops as the local velocity peaks to $\sim 100\%$ above the mean front velocity. Examination of the fracture surface suggests that the release of curvature and the velocity jump are coincident with the death of the microbranch. Immediately afterwards (**4**) the velocity drops again as a new microbranch nucleates. In contrast to the strong fluctuations along **A**, the “quiet” adjacent mirrorlike region [25] marked **B** shows practically no change in curvature. We do, however, see significant variations in velocity which, after a delay, follow the changes in velocity along the branch line **A**.

Let us consider these results. First, the association of microbranch nucleation and death with local velocity decrease and increase along the branch line is qualitatively consistent with *local* energy balance: $G = \Gamma$. Our second observation is that after the front acquires curvature and the microbranch dies, the local velocity does not only return to the mean velocity level, but doubles it. These observations suggest that front curvature must be increasing G locally to produce these high velocity peaks. This idea is supported by a well-known result derived by Rice [26] for *static* fronts. Rice computed the *static* stress contribution made by a small in-plane perturbation to a straight front. When a front $x = x_0 + \delta x(z)$ is perturbed in the sense that $|dx/dz| \ll 1$, the stress maintains the inverse-square-root singularity at each point along the front, but $K = K_0 + \delta K(z)$ varies along the front as

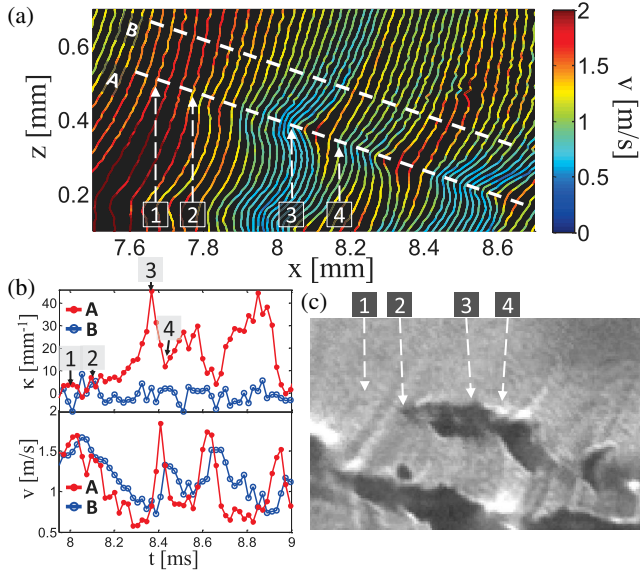


FIG. 2 (color online). Three consecutive branching events along a branch line. (a) The series of fronts captured during branch-line formation. Colors indicate local velocity levels. Marked points show (1) an unperturbed front, (2) local velocity decrease associated with microbranch nucleation, (3) maximum local curvature, and (4) formation of a new microbranch. (b) Plots of local curvature (top) and local velocity (bottom) along the cuts **A** (dots) and **B** (open symbols) in (a). (c) The corresponding post mortem showing the branch line associated with the fronts in (a).

$$\frac{\delta K(z)}{K_0} = \frac{1}{2\pi} PV \int \frac{\delta x(z') - \delta x(z)}{(z' - z)^2} dz', \quad (1)$$

where xz is the fracture plane, x is the crack propagation direction, and the unperturbed front is $x(z) = x_0$ with a stress intensity factor K_0 . A salient feature of this integral is that it is solely determined by the front geometry. It generates a stabilizing force (“line tension”) that tends to restore the front to a flat configuration, since $G \propto K^2$ or $\delta G/G_0 = 2\delta K/K_0$. This force is countered by the local increase of the fracture energy $\delta\Gamma(z)$, generated by the microbranch. In general, we cannot measure $\delta\Gamma$ directly. When, however, a microbranch “dies,” $\delta\Gamma \sim 0$ and the elastic energy stored in the front curvature locally accelerates the crack. Evidence for this is seen in Fig. 2(b) along the **A** line, where we see that each peak in curvature is followed by a peak in local front velocity as the line tension is released.

While these statements appear qualitatively accurate, are they quantitatively correct? Let us now focus on the moment of microbranch death, the moments corresponding to the peak in local front velocity at the instant where $\delta\Gamma \sim 0$. We consider thirteen cases, with mean crack velocities ranging between 0.1–0.3 c_R , where we identify a clear transition from a branched to smooth surface in the *post-mortem* image.

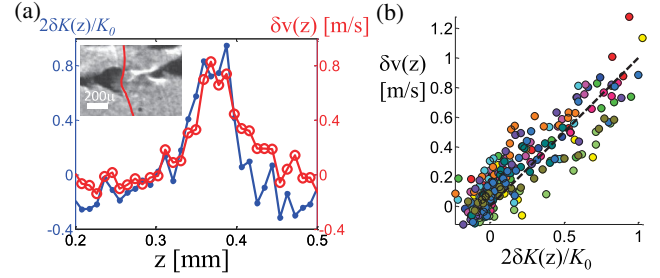


FIG. 3 (color online). Comparison of the instantaneous velocity and local stress intensity factor, computed by Eq. (1), along a front at the moment of microbranch death. (a) Spatial stress (dots) and velocity variations (open circles) for the front in the inset are well correlated. (inset) Overlay of the front at the moment of maximal velocity on the resulting post-mortem fracture surface. Front positioning is approximate. (b) $\delta v(z)$ vs $\delta G/G_0 = 2\delta K/K_0$ for both the data in (a) and twelve such releases from other events. Different colors indicate different events. The dashed $y = x$ line is a guide to the eye.

At these instances we can directly compare the relative velocity deviation from the mean $\delta v(z) = v(z) - \langle v(z) \rangle$ and the built-up energy release rate $\delta G/G_0 = 2\delta K/K_0$ computed by Eq. (1). As Eq. (1) was derived for fronts parallel to the z axis, we computed $\delta K/K_0$ over symmetric regions of the front around the point of maximum curvature by first rotating the front to make edges of this region sit on an $x = \text{const}$ line. The average $\langle v(z) \rangle$ was computed for velocity values in the same region.

The result of the comparison is presented in Fig. 3. We see that the correction for the stress intensity factor given by Eq. (1) is clearly correlated with the instantaneous velocity profile at the moment of microbranch death. Despite being only a first-order correction, Eq. (1) is shown here to work quite well. This is demonstrated locally along a typical front in Fig. 3(a) and in another twelve similar instances in Fig. 3(b). It is a manifest demonstration of how front curvature is translated into motion. We note that we see no observable effect of the proximity of neighboring branch lines in this collapse. This is consistent with the observations of Ref. [27] in quasistatic propagation. Since we compare here a dimensionless quantity with velocity $\delta v = 2v_0\delta K/K_0$, the coefficient of proportionality v_0 must also have the dimensions of velocity. Its values $v_0 = 1 \pm 0.3$ m/s are far from the characteristic wave velocity ($c_R \sim 5$ m/s), but close to the measured mean crack velocities $\langle v \rangle = 0.7$ – 1.5 m/s. This might be related to the strong velocity dependence of the fracture energy in polyacrylamide gels [4]. We give a possible origin of this coefficient in the Supplemental Material [28,29].

Let us now consider the front evolution depicted in Fig. 4, where the process leading to the buildup of the line tension prior to the death of a microbranch is highlighted. During the buildup of the line tension, v along the crest of the branch line is approximately constant during the rapid

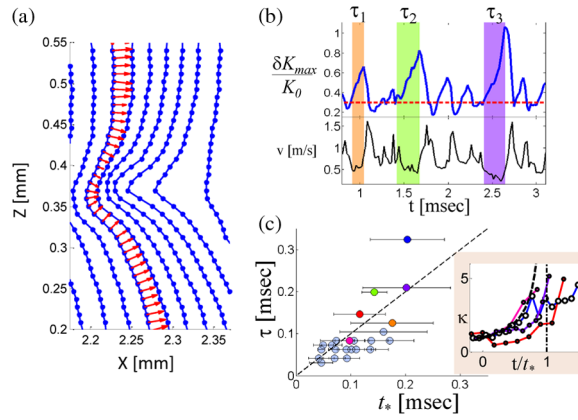


FIG. 4 (color online). (a) Cusp formation during a typical microbranching event. We present every second front. Red arrows denote normal displacements. (b) Time series of the peak value $\delta K/K_0$ (top) and front velocity (bottom) along the crest of a branch line show a few cycles of buildup and release. We define the microbranch lifetime τ by the interval over which $\delta K/K_0$ first surpasses (microbranch birth) and then drops below (microbranch death) a threshold value 0.3 (red line). Note that during stress buildup, the local velocity stays approximately constant. Shaded areas: 3 examples of τ for different branching events. (c) τ compared to predicted cusp formation times $t_* = 1/(\kappa_0 v_n)$. κ_0 is the initial maximal curvature and v_n is the initial normal velocity at the location of maximum curvature. Colored points denote events in (b) and those shown in the inset. (inset) Curvature blowup dynamics. Time is normalized by t_* . $t = 0$ corresponds to the initial front. κ is the maximal curvature of each successive front normalized by κ_0 . Black dashed line: predicted finite-time blowup.

increase in curvature that leads to cusp formation. In Fig. 4(a) we present a series of fronts leading to a release event. The release is preceded by a rapid increase in curvature which culminates when the front attains a cusplike form. As in the events depicted in Fig. 2, local increases in v immediately follow cusp formation.

It is well known that propagating fronts may develop shocks in curvature [30], or cusps. If a cusp does form, we would expect the stress intensity factor to locally diverge according to Eq. (1). It is, therefore, likely that this high stress could promote the death of the microbranch and hence trigger the local release of a front.

How can this picture be tested? A nearly constant normal velocity, $v(z, t) = v_n$, where v_n is taken to be the initial velocity at the beginning of the event, coupled with an initially curved section of the front will spontaneously produce a cusp with no further assumptions. To see this, we rewrite $v = v_n$ in terms of front coordinates: $\partial_t x = v_n \sqrt{1 + (\partial_z x)^2}$. Differentiating by z , we obtain an equation of motion for the local slope $u = -\partial_z x$:

$$\frac{\partial u}{\partial t} + v_n \frac{u}{\sqrt{1 + u^2}} \frac{\partial u}{\partial z} = 0. \quad (2)$$

It is known that this equation develops shocks (i.e., cusps in u) in finite time, where $\partial_z^2 x = -\partial_z u \sim (t_* - t)^{-1}$ and $t_* = 1/(\kappa_0 v_n)$ when κ_0 is the maximal front curvature at $t = 0$.

The constant velocity model provides us with a prediction for the time required for the front to form a cusp t_* . Is this time scale related to the lifetime τ of a microbranching event? To test this, we consider an ensemble of microbranching events. As depicted in Fig. 4(b), we define τ as the time between the initial stress buildup and stress release points for each event, using a threshold of $\delta K/K_0 = 0.3$ to define these points. We use the initial maximal curvature κ_0 and the normal velocity v_n to evaluate the predicted cusp formation time t_* . (We measure κ_0 by rotating the front to eliminate any mean slope and fitting the region of highest curvature with a parabola. v_n is the average normal velocity over the same range in z .) We consider only events where the estimated error in t_* was no larger than 50%.

As shown in Fig. 4(c) this simplistic model works rather well. Within the limits of our experimental resolution $\tau \approx t_*$; over the stress buildup we observe rapid growth of curvature that culminates near the predicted t_* . Moreover, our predicted t_* indeed correspond to the death of each microbranch.

This being said, microbranch dynamics do not always proceed directly to curvature blowup. In some cases the front may “slip”; undergoing partial release during an event before final cusp formation. An example of such slip is the curve denoted by open symbols in the inset of Fig. 4(c).

In conclusion, we have presented the first experimental exploration of rapid crack front in-plane dynamics. Our observation that microbranches act as sinks that effectively increase the fracture energy agrees qualitatively with a local energy balance. The induced front curvature may explain the microbranch localization in z as the velocity overshoots upon microbranch release appear to generate new microbranches, thereby forming branch lines. Moreover, we have shown that the pronounced velocity overshoot at the moment of microbranch death is well correlated with the static contribution for the stress intensity factor resulting from front curvature. This is a clear demonstration of how stresses distributed along a crack front translate locally into dynamics.

Nevertheless, the demonstrated validity of the expression [Eq. (1)] is intriguing for several reasons. First, Eq. (1) is a first-order correction in the local slope of the front, while the fronts we considered contained mild slopes. Our current data do not have the precision to investigate the effects of higher order corrections [31–33] and it would be interesting to test their implications in the future. Second, Eq. (1) was derived for static fronts while we consider moving fronts. In the work of Ramanathan and Fisher [10] the static theory was expanded to include dynamic effects. These include waves that propagate on the front at velocity $\sim \sqrt{c_R^2 - v^2}$.

Since in our experiments $v^2/c_R^2 \sim 0.1 \ll 1$ and because of the small size of microbranches ($\sim 100 \mu$) and relatively long lifetime ($\sim 100 \mu s$), we expect any signature of the waves to average during microbranch growth. We might expect the agreement with Eq. (1) to break down at larger v/c_R , when such inertial effects become important. Last, apart from the increase in fracture area caused by the out-of-plane motions of the front, microbranches should also influence the local energy release rate by introducing shearing stresses. The combined effects of these contributions on dynamics demand further study.

In addition, while recent experiments [34] provide a mechanism for microbranch birth, we have seen that the front dynamics induced by this “birth” could lead to eventual microbranch death; front curvature caused by microbranch initiation may lead to microbranch death via the large line tension generated by dynamic cusp formation. This scenario could provide a key to understanding why microbranches remain small and do not develop. The above results are all purely 3D effects that underline the necessity of extending fracture mechanics to the third dimension.

The authors acknowledge the support of the European Research Council (Grant No. 267256) and Israel Science Foundation (Grant No. 76/11). We thank Baruch Meerson for helpful discussions on cusp formation in fronts.

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- [1] L. B. Freund, *Dynamic Fracture Mechanics* (Cambridge University Press, Cambridge; New York, 1990).
- [2] E. Bouchbinder, J. Fineberg, and M. Marder, *Annu. Rev. Condens. Matter Phys.* **1**, 371 (2010).
- [3] E. Sharon and J. Fineberg, *Nature (London)* **397**, 333 (1999).
- [4] T. Goldman, A. Livne, and J. Fineberg, *Phys. Rev. Lett.* **104**, 114301 (2010).
- [5] J. Fineberg and M. Marder, *Phys. Rep.* **313**, 1 (1999).
- [6] D. Hull, *Fractography: Observing, Measuring and Interpreting Fracture Surface Topography* (Cambridge University Press, Cambridge, England, 1999).
- [7] O. Ronsin, C. Caroli, and T. Baumberger, *Europhys. Lett.* **105**, 34001 (2014).
- [8] J. Scheibert, C. Guerra, F. Célarié, D. Dalmas, and D. Bonamy, *Phys. Rev. Lett.* **104**, 045501 (2010).
- [9] C. Guerra, J. Scheibert, D. Bonamy, and D. Dalmas, *Proc. Natl. Acad. Sci. U.S.A.* **109**, 390 (2012).
- [10] S. Ramanathan and D. S. Fisher, *Phys. Rev. Lett.* **79**, 877 (1997).
- [11] J. W. Morrissey and J. R. Rice, *J. Mech. Phys. Solids* **48**, 1229 (2000).
- [12] J.-B. Leblond, A. Karma, and V. Lazarus, *J. Mech. Phys. Solids* **59**, 1872 (2011).
- [13] M. Adda-Bedia, R. E. Arias, E. Bouchbinder, and E. Katzav, *Phys. Rev. Lett.* **110**, 014302 (2013).
- [14] A. J. Pons and A. Karma, *Nature (London)* **464**, 85 (2010).
- [15] D. Dalmas, E. Barthel, and D. Vandembroucq, *J. Mech. Phys. Solids* **57**, 446 (2009).
- [16] J. Chopin, A. Prevost, A. Boudaoud, and M. Adda-Bedia, *Phys. Rev. Lett.* **107**, 144301 (2011).
- [17] S. Patinet, L. Alzate, E. Barthel, D. Dalmas, D. Vandembroucq, and V. Lazarus, *J. Mech. Phys. Solids* **61**, 311 (2013).
- [18] S. Xia, L. Ponson, G. Ravichandran, and K. Bhattacharya, *Phys. Rev. Lett.* **108**, 196101 (2012).
- [19] S. Santucci, M. Grob, R. Toussaint, J. Schmittbuhl, A. Hansen, and K. J. Måløy, *Europhys. Lett.* **92**, 44001 (2010).
- [20] D. Bonamy, S. Santucci, and L. Ponson, *Phys. Rev. Lett.* **101**, 045501 (2008).
- [21] S. Patinet, D. Vandembroucq, and S. Roux, *Phys. Rev. Lett.* **110**, 165507 (2013).
- [22] E. Sharon, G. Cohen, and J. Fineberg, *Phys. Rev. Lett.* **88**, 085503 (2002).
- [23] A. Livne, G. Cohen, and J. Fineberg, *Phys. Rev. Lett.* **94**, 224301 (2005).
- [24] E. Sharon and J. Fineberg, *Phys. Rev. B* **54**, 7128 (1996).
- [25] The fracture surfaces observed here are truly mirrorlike, no parabolic markings due to void formation (such as those observed in [9]) are in evidence.
- [26] J. Rice, *J. Appl. Mech.* **52**, 571 (1985).
- [27] J. Chopin, Theses, Université Pierre et Marie Curie - Paris VI, 2010.
- [28] See Supplemental Material at <http://link.aps.org/supplemental/10.1103/PhysRevLett.114.175501> for (i) heuristic derivation of the coefficient v_0 defined in the expression $\delta v = 2v_0\delta K/k_0$ and (ii) evidence for the constancy of v during microbranch growth that was assumed in the derivation of Eq. (2), which contains Ref. [29].
- [29] T. Goldman, “Measurements of Fracture Energy” (unpublished).
- [30] M. Kardar, G. Parisi, and Y.-C. Zhang, *Phys. Rev. Lett.* **56**, 889 (1986).
- [31] M. Adda-Bedia, E. Katzav, and D. Vandembroucq, *Phys. Rev. E* **73**, 035106 (2006).
- [32] J.-B. Leblond, S. Patinet, J. Frelat, and V. Lazarus, *Eng. Fract. Mech.* **90**, 129 (2012).
- [33] M. Vasoya, V. Lazarus, and L. Ponson, *Procedia Mater. Sci.* **3**, 2142 (2014).
- [34] T. G. Boué, G. Cohen, and J. Fineberg, *Phys. Rev. Lett.* **114**, 054301 (2015).